

Hybrid DFT with numeric atomic orbitals

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Florian Knoop (florian.knoop@liu.se)



DFT in a nutshell

Hohenberg-Kohn [1]

Ground state properties of matter determined by density



Walter Kohn
(1923-2016)

$$\Psi_{\text{el}}(r_1, r_2, \dots, r_{3N}) \iff \rho(x, y, z)$$

- **Reduce degrees of freedom** by a factor N ($\sim 10^{23}$)
- **Electron ground state properties:** total energy, forces, dipole moments, ...

[1] P. Hohenberg, W. Kohn, *Phys Rev.* **136**, B864 (1964)

[2] W. Kohn, L.J. Sham, *Phys. Rev.* **140**, A1133 (1965)

DFT in a nutshell

Kohn-Sham [2]

Map electron density on effective one-particle orbitals

$$\rho(r) \implies \sum_i f_i \left| \psi_i^{\text{KS}} \right|^2$$

- can be solved **self-consistently** via **Schrödinger-like equation**
- Kohn-Sham orbitals are **not electrons!**



Walter Kohn
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[1] P. Hohenberg, W. Kohn, *Phys Rev.* **136**, B864 (1964)

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KS equations:

$$\left(-\frac{1}{2} \nabla^2 + V^{\text{KS}} \left[\{ \psi_i^{\text{KS}} \} \right] \right) \psi_i^{\text{KS}} = \epsilon_i \psi_i^{\text{KS}}$$


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DFT energy

→ DFA (approximation)

$$E_{\text{total}} = T_0 + \int \rho V_{\text{nuc}} d^3\mathbf{r} + \frac{1}{2} \iint \frac{\rho(\mathbf{r}_1)\rho(\mathbf{r}_2)}{r_{12}} d^3\mathbf{r}_1 d^3\mathbf{r}_2 + E_{\text{XC}}, \quad (3)$$

Schrödinger-like equation

[1] P. Hohenberg, W. Kohn, *Phys Rev.* **136**, B864 (1964)

[2] W. Kohn, L.J. Sham, *Phys. Rev.* **140**, A1133 (1965)

DFT: Basissets

used to represent KS states

Basisset: Complete set of functions to represent KS states

$$\psi_i^{\text{KS}}(\mathbf{r}) = \lim_{K \rightarrow \infty} \sum_{k=1}^K c_k \phi_k(\mathbf{r})$$

Problem

Complete basis set = infinite size ($K = \infty$)

Solution

Find *good enough* basis ($K < \infty$)

Popular basissets

1) Plane waves (VASP, Quantum Espresso, ...)

basically a Fourier transform, K corresponds to highest energy: $|\mathbf{k}| \leq K = E_{\text{cut}}/\hbar$

$$\phi_{\mathbf{k}}(\mathbf{r}) \propto \exp(i\mathbf{k} \cdot \mathbf{r})$$

Advantages 👍

easy integrals, natural choice for periodic systems, single cutoff

Disadvantages 🙅

huge basis needed, cores need extra treatment (pseudopotentials, PAW, ...)

non-periodic systems

Popular basissets

2) Gaussian type orbitals (Gaussian, ORCA, ...)

$$\phi_{k \equiv lmn; \alpha}(\mathbf{r}) \propto x^l y^m z^n \exp(-\alpha r^2)$$

Advantages 👍

analytic integrals

local → good for molecules

Disadvantages 👎

more cutoff parameters

Popular basis sets

3) Numeric atom-centered orbitals (FHI-aims, DFTB+, ...)

$$\phi_{k \equiv ilm}(\mathbf{r}) \propto \frac{u_i(r)}{r} Y_{lm}(\varphi, \vartheta)$$

Advantages 👍

local → good for molecules, **treat molecules and solids on same footing**

small amount of basis functions needed → **linear scaling**

all electrons can be treated with little extra cost (no pseudos or PAW)

Disadvantages 👎

movable grids → complicated Pulay terms

more difficult to handle analytically than plane waves or Gaussians

Popular basissets

3) Numeric atomic orbitals (e.g. PAW, TB+, ...)

Question:

Can I treat periodic systems with a local basisset?

Answer:

Yes.

Advantages

local

small and

all electrons

Disadvantages

movable grid, complicated Pulay terms

more difficult to handle analytically than plane waves or Gaussians

NAOs in FHI-aims

What should we know?

light

Get to know a system, geometry optimization, molecular dynamics, ...

intermediate

tighter grids, often converged

tight

Converged geometries and energies for checking

really_tight

Overcomplete Benchmark basisset for double checking

NAOs in FHI-aims

What should we know?

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Get to know a system, geomet

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Overcomplete Benchmark basi



lar dynamics, ...

ing

NAOs in FHI-aims

Example: Silicon

$$\phi_{k \equiv ilm}(\mathbf{r}) \propto \frac{u_i(r)}{r} Y_{lm}(\varphi, \vartheta)$$

Si, fcc diamond	light	intermediate	tight	really_tight
r_{cut} [Å]	5	6	6	6
# Integration points	11,208	35,840	35,840	56,860
Multipole l_{max}	4	6	6	8
Max. # basis functions	36	68	78	78
Error [meV per atom]	28.1	11.6	0.6	0.0
Time per scf step [s] [*]	0.4	3.4	4.7	9.8

OK fine – why bother?

Efficient All-electron Hybrid Density Functionals for Atomistic Simulations Beyond 10,000 Atoms

Sebastian Kokott,¹ Florian Merz,² Yi Yao,³ Christian Carbogno,¹ Mariana Rossi,⁴ Ville Havu,⁵ Markus Rampp,⁶ Matthias Scheffler,¹ and Volker Blum^{3,7}

¹*The NOMAD Laboratory at the Fritz Haber Institute of the Max-Planck-Gesellschaft and IRIS Adlershof of the Humboldt-Universität zu Berlin, Germany*

²*Lenovo HPC Innovation Center, Stuttgart, Germany*

³*Thomas Lord Department of Mechanical Engineering and Material Science, Duke University, Durham, North Carolina 27708, USA*

⁴*MPI for the Structure and Dynamics of Matter, Luruper Chaussee 149, 22761 Hamburg, Germany*

⁵*Department of Applied Physics, School of Science, Aalto University, Espoo, Finland*

⁶*Max Planck Computing and Data Facility, 85748 Garching, Germany*

⁷*Department of Chemistry, Duke University, Durham, North Carolina 27708, USA*

(Dated: 18 March 2024)

Hybrid density functional approximations (DFAs) offer compelling accuracy for *ab initio* electronic-structure simulations of molecules, nanosystems, and bulk materials, addressing some deficiencies of computationally cheaper, frequently used semilocal DFAs. However, the computational bottleneck of hybrid DFAs is the evaluation of the non-local exact exchange contribution, which is the limiting factor for the application of the method for large-scale simulations. In this work, we present a drastically optimized resolution-of-identity-based real-space implementation of the exact exchange evaluation for both non-periodic and periodic boundary conditions in the all-electron code FHI-aims, targeting high-performance CPU compute clusters. The introduction of several new refined Message Passing Interface (MPI) parallelization layers and shared memory arrays according to the MPI-3 standard were the key components of the optimization. We demonstrate significant improvements of memory and performance efficiency, scalability, and workload distribution, extending the reach of hybrid DFAs to simulation sizes beyond ten thousand atoms. As a necessary byproduct of this work, other code parts in FHI-aims have been optimized as well, e.g., the computation of the Hartree potential and the evaluation of the force and stress components. We benchmark the performance and scaling of the hybrid DFA based simulations for a broad range of chemical systems, including hybrid organic-inorganic perovskites, organic crystals and ice crystals with up to 30,576 atoms (101,920 electrons described by 244,608 basis functions).

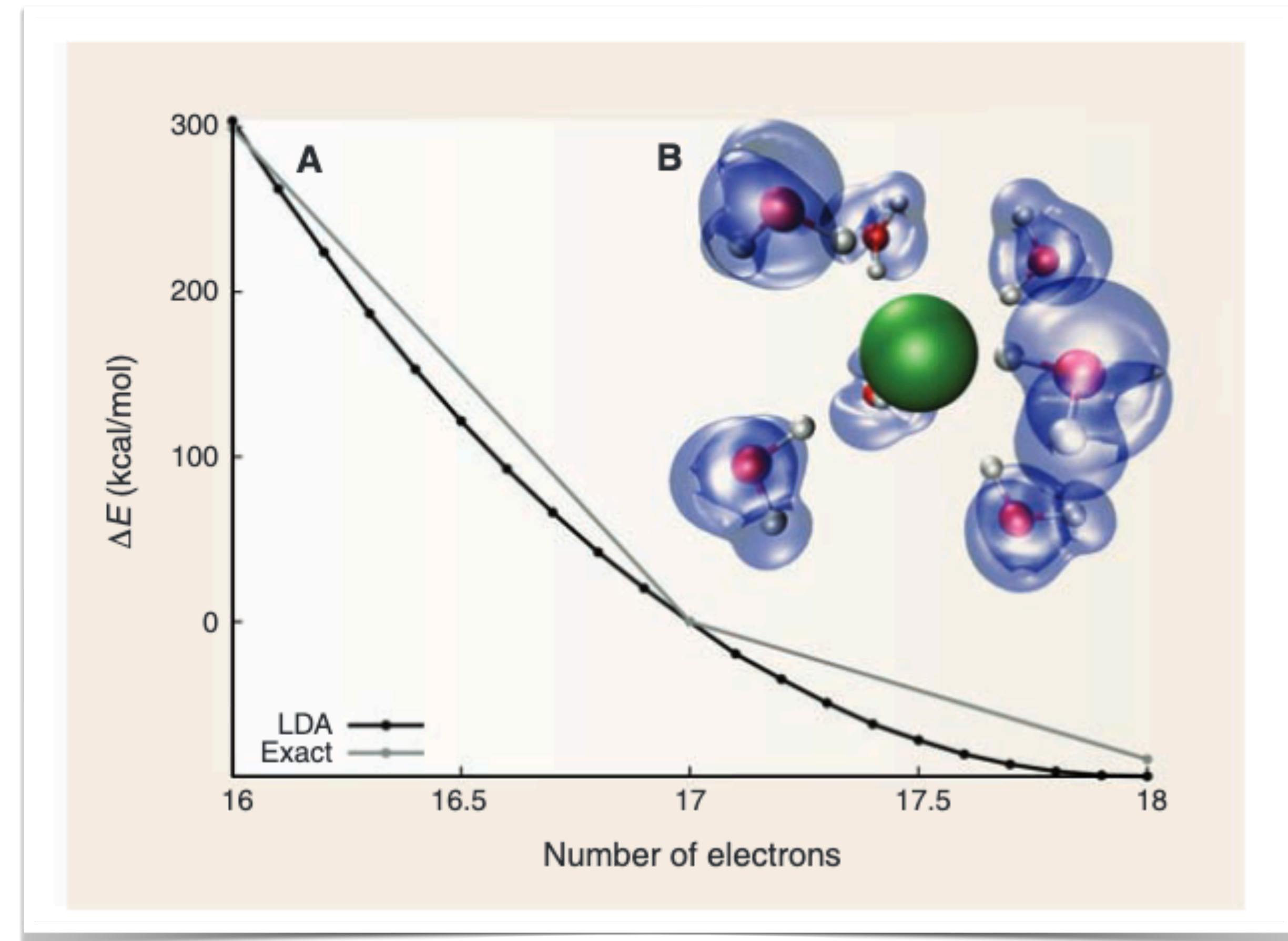
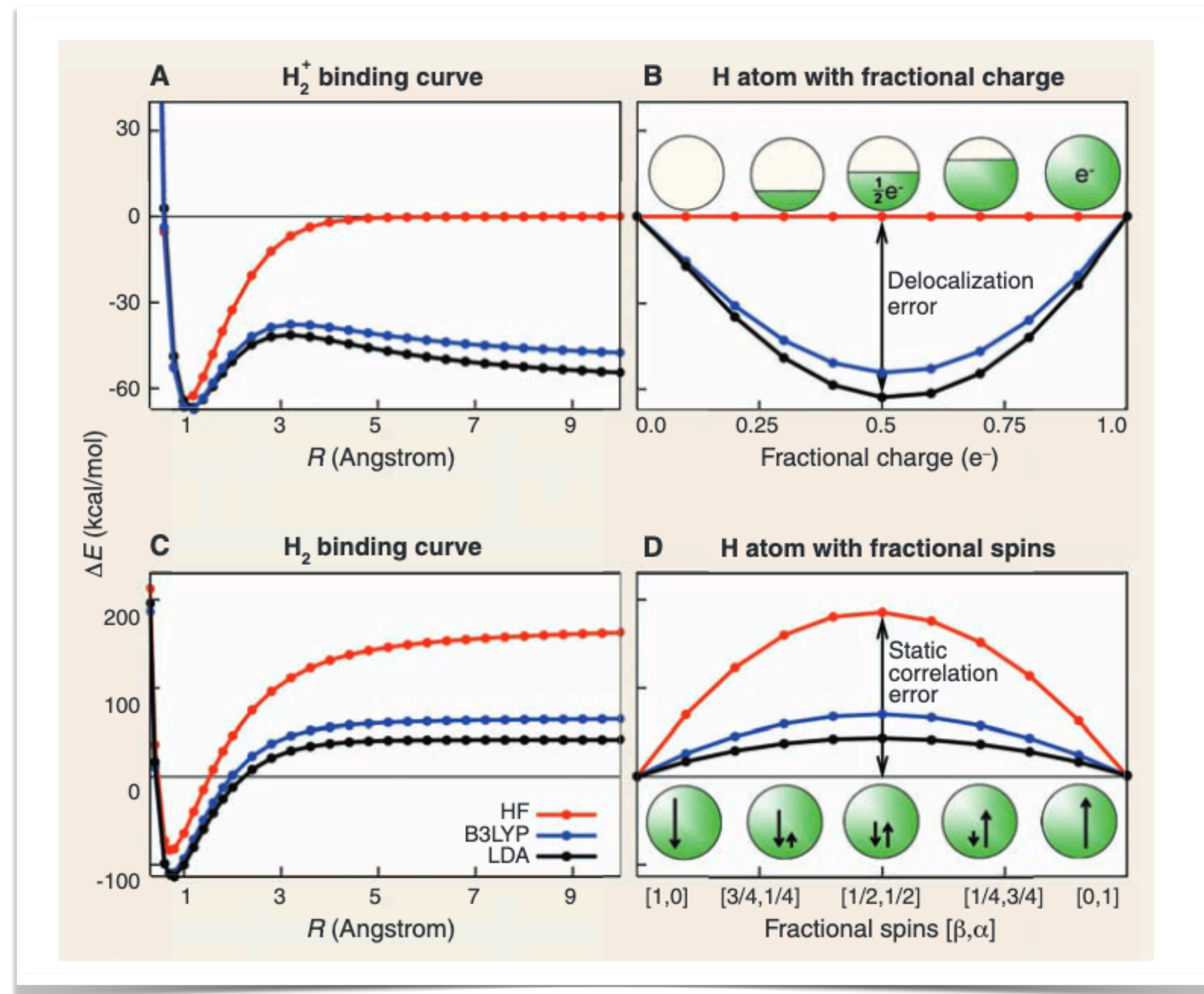
15 Mar 2024

[cond-mat.mtrl-sci]

Interlude: Hybrid functionals

Why hybrid functionals?

(semi-local) DFT can fail dramatically, e.g., **delocalization error** [1]



Why hybrid functionals?

exchange can be computed exactly → **exact exchange (EXX or HF) [1]**

$$E_{\text{total}} = T_0 + \int \rho V_{\text{nuc}} d^3\mathbf{r} + \frac{1}{2} \iint \frac{\rho(\mathbf{r}_1)\rho(\mathbf{r}_2)}{r_{12}} d^3\mathbf{r}_1 d^3\mathbf{r}_2 + E_{\text{XC}}, \quad (3)$$

Naïve:

$$E_{\text{XC}} = E_X + E_C^{\text{LSDA}}, \quad (8)$$

Becke: Use adiabatic connection from noninteracting ($\lambda=0$) to interacting ($\lambda=1$)

$$E_{\text{XC}} = \int_0^1 U_{\text{XC}}^\lambda d\lambda, \quad (9)$$

some approximations

$$E_{\text{XC}} \approx \frac{1}{2} E_X + \frac{1}{2} U_{\text{XC}}^{\text{LSDA}}, \quad (12)$$

How to get a good hybrid?

About 25% exact exchange is good [1]

$$E_{\text{xc}}^{\text{hyb}} = E_{\text{xc}}^{\text{DFA}} + a_0(E_{\text{x}} - E_{\text{x}}^{\text{DFA}}), \quad (5)$$

Perdew, Ernzerhof, Burke (**PBE0**) [1]:

“We will show that

$a_0 \approx 0.25$ is to be expected for the atomization energies of most molecules,

but also that

larger values of a_0 may be more appropriate for the total energies of atoms and molecules,
and

smaller values for atomization energies of molecules with nearly degenerate ground states
for the unperturbed ($\lambda = 0$) problem.”

[1] J. P. Perdew, M. Ernzerhof, and K. Burke, *J. Chem. Phys.* **105**, 9982 (1996).

Problems with exact exchange (EXX, HF)

$$E_x = \sum_{\mu\nu,\lambda\sigma} c_{\mu\nu,\lambda\sigma} \langle \mu\nu | \lambda\sigma \rangle$$

$$\langle \mu\nu | \lambda\sigma \rangle = \iint d\mathbf{r}_1 d\mathbf{r}_2 \psi_\mu(\mathbf{r}_1) \psi_\nu(\mathbf{r}_1) \frac{1}{r_{12}} \psi_\lambda(\mathbf{r}_2) \psi_\sigma(\mathbf{r}_2)$$

↑
two-electron Coulomb integrals

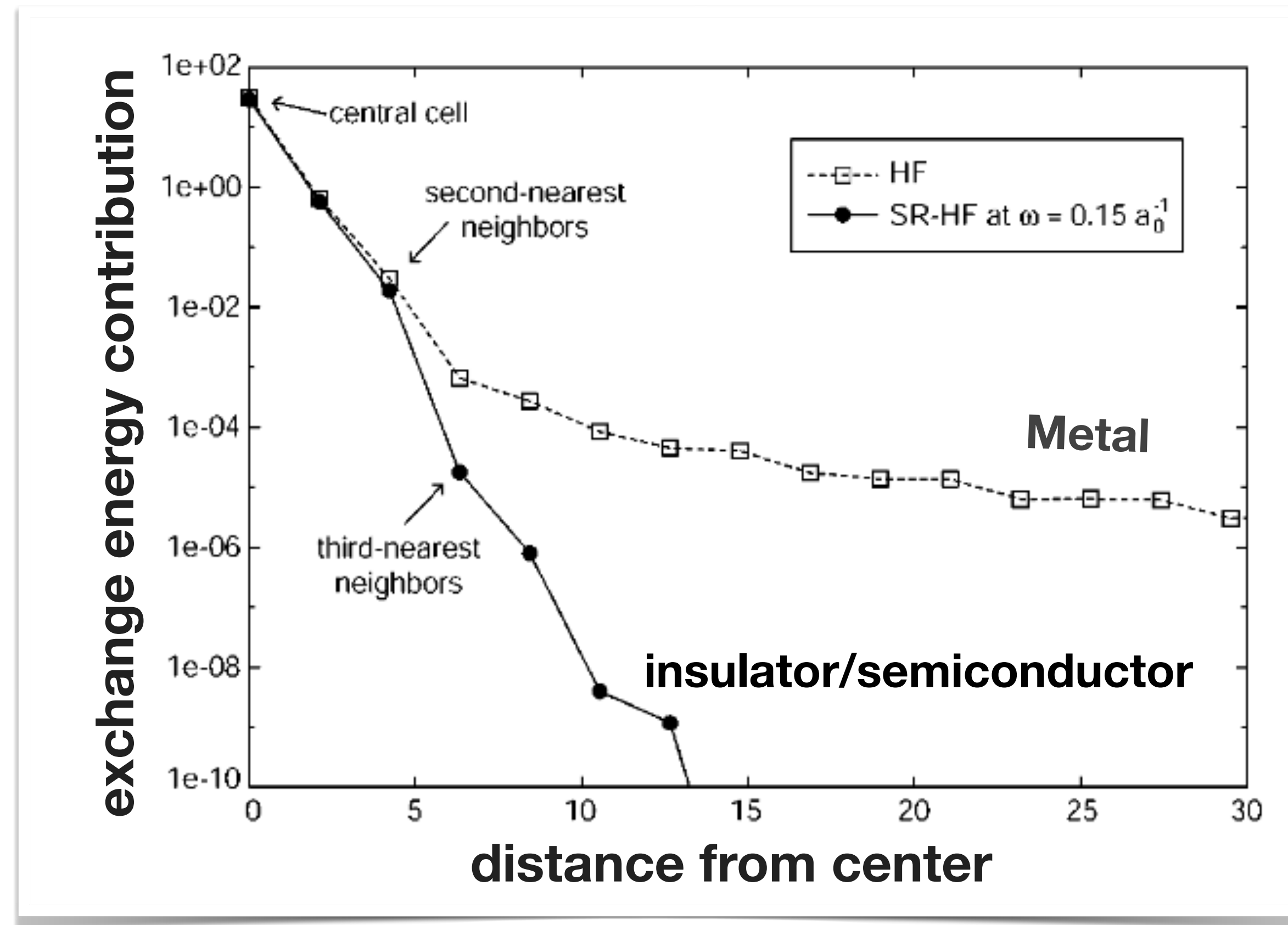
↑
Coulomb interaction

Problem 1: Coulomb interaction is long-ranged (= **expensive**)

Problem 2: (unscreened) Coulomb interactions **diverges for metals**

Solve problem 1: Nearsightedness

Range of exchange decays exponentially for systems with gap [2, 3]



[1] J. Heyd, G. E. Scuseria, and M. Ernzerhof, J Chem Phys **118**, 8207 (2003).

[2] E. Prodan and W. Kohn, Proc. Natl. Acad. Sci. **102**, 11635 (2005).

[3] W. Kohn, Int. J. Quantum Chem. **56**, 229 (1995).

Introduce range separation in HF computation

→ HSE06 [1, 2]

$$\frac{1}{r} = \frac{\text{erfc}(\omega r)}{r} + \frac{\text{erf}(\omega r)}{r}$$

short-range part

neglect long-range part

$$\langle \mu\nu | \lambda\sigma \rangle^{\text{SR}} = \iint dr_1 dr_2 \psi_\mu(\mathbf{r}_1) \psi_\nu(\mathbf{r}_1) \frac{\text{erfc}(\omega r_{12})}{r_{12}} \psi_\lambda(\mathbf{r}_2) \psi_\sigma(\mathbf{r}_2)$$

$$\langle \mu\nu | \lambda\sigma \rangle^{\text{LR}} \approx 0$$

$$E_{\text{xc}}^{\text{HSE}} = aE_x^{\text{HF,SR}}(\omega) + (1-a)E_x^{\text{PBE,SR}}(\omega) + E_x^{\text{PBE,LR}}(\omega) + E_c^{\text{PBE}}$$

[1] J. Heyd, G. E. Scuseria, and M. Ernzerhof, J Chem Phys **118**, 8207 (2003).

[2] J. Heyd, G. E. Scuseria, and M. Ernzerhof, J Chem Phys **124**, 219906 (2006).

Introduce range-separated hybrid computation

computation

Bottom line 1:

→ HSE06 [1]

Hybrid functionals [1, 2] solve *some* errors [3]

of (semi-)local DFAs.

Bottom line 2:

Range-separated hybrid functionals

can do the same but *faster and for metals*.

[1] S. Kümmel and L. Kronik, Rev Mod Phys **80**, 3 (2008)

[2] A. M. Teale *et al.*, Phys Chem Chem Phys (2022)

[3] P. Borlido *et al.*, J. Chem. Theory Comput. **15**, 5069 (2019)

$\langle \mu\nu | \lambda\sigma \rangle^{SR}$

$\langle \mu\nu | \lambda\sigma \rangle^{LR} \approx 0$

$E_c^{PBE}(\omega) + E_c^{PBE}$

[1] J. Heyd, G. E. Scuseria, and M. Ernzerhof, J Chem Phys **118**, 8207 (2003).

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Back to FHI-aims

Efficient All-electron Hybrid Density Functionals for Atomistic Simulations Beyond 10,000 Atoms

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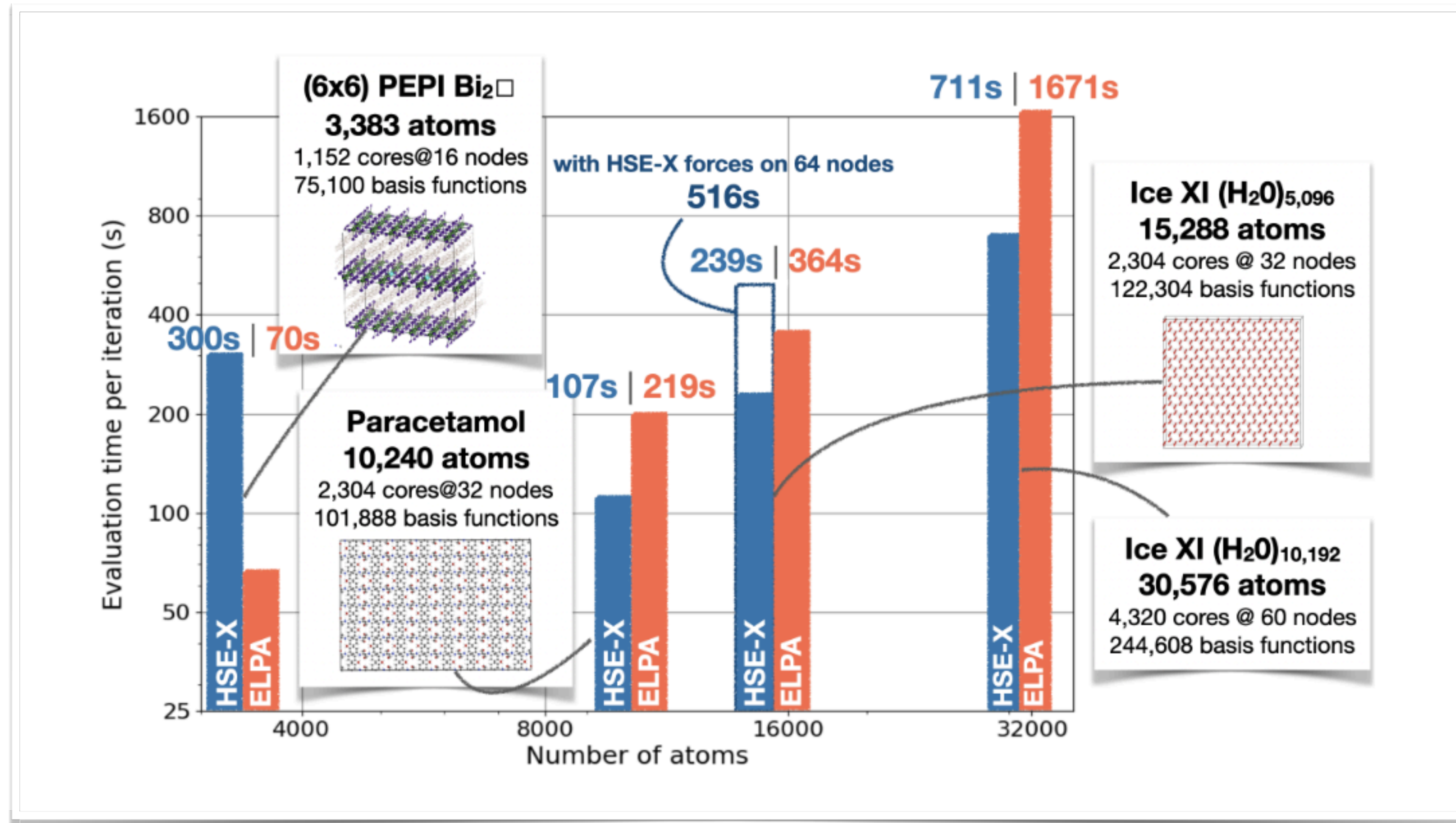
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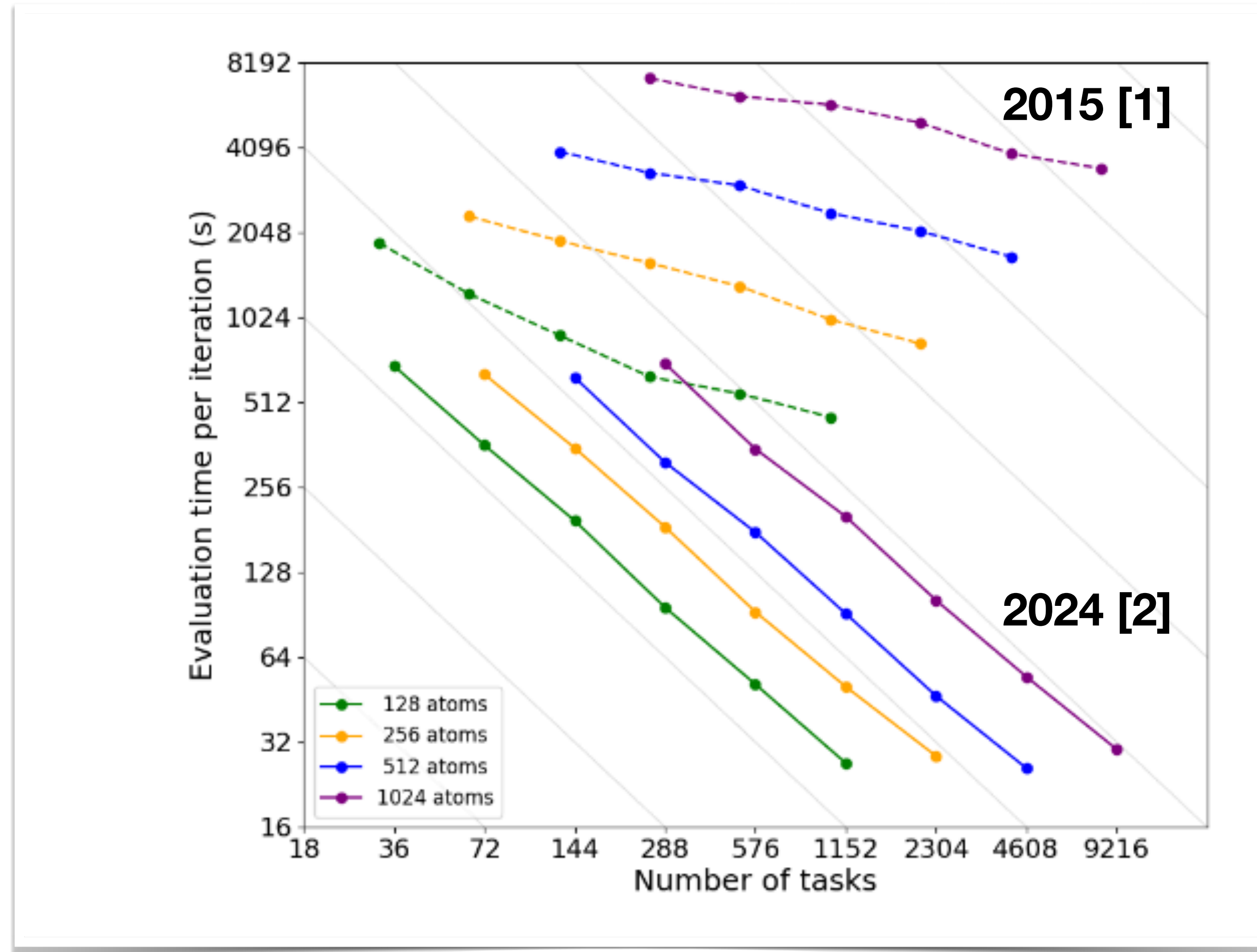
15 Mar 2024

[cond-mat.mtrl-sci]

HSE06 for *many* atoms



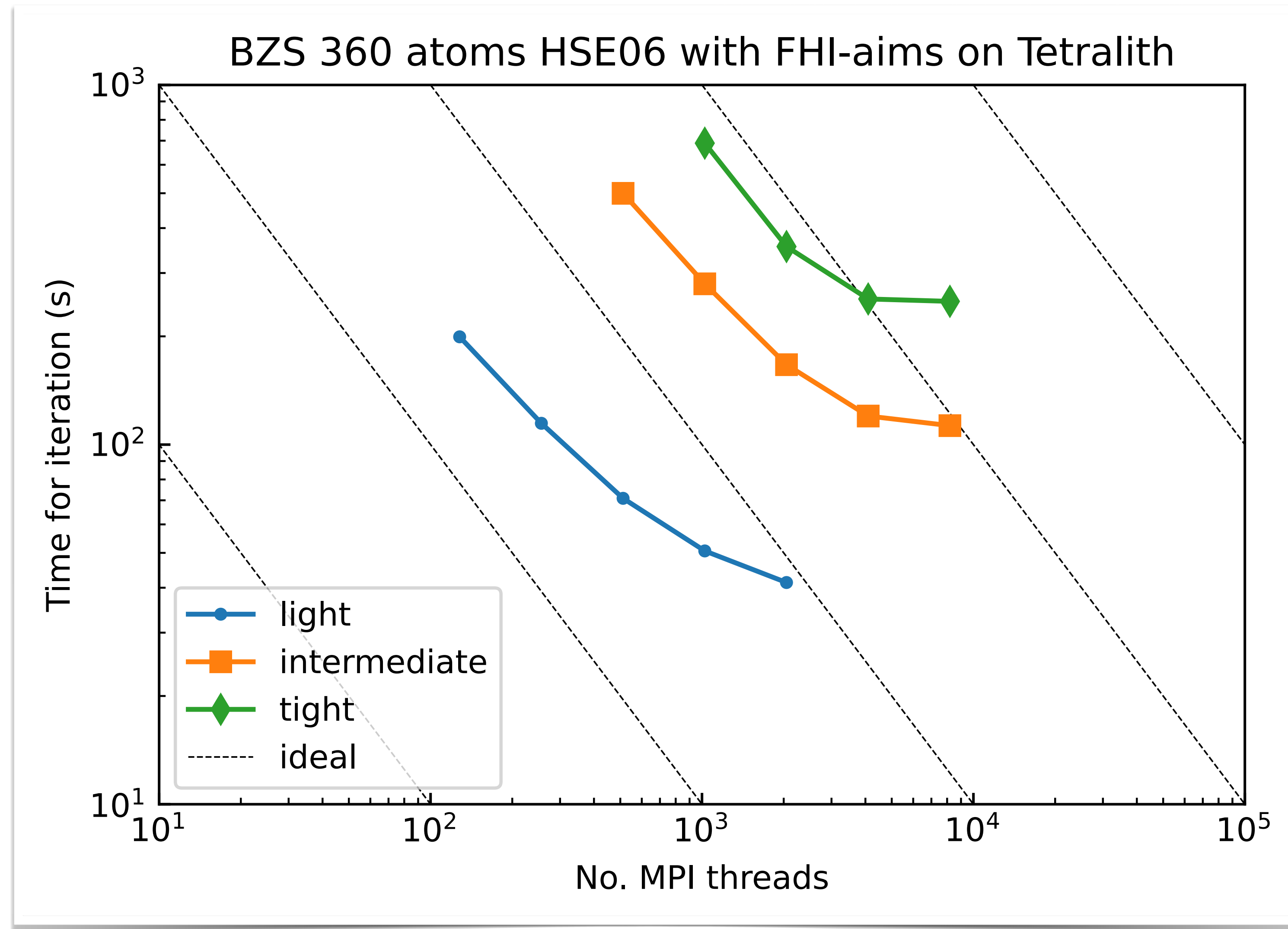
Strong scaling



[1] S. V. Levchenko *et al.*, Comput Phys Commun **192**, 60 (2015)

[2] S. Kokott *et al.*, *arXiv* 2403.10343 (2024)

Strong scaling on NSC's tetralith



Strong scaling on NCS's tetralith

Dardel?

Unfortunately no. Cray MPI libraries are too buggy 🙄.

PDC is aware.

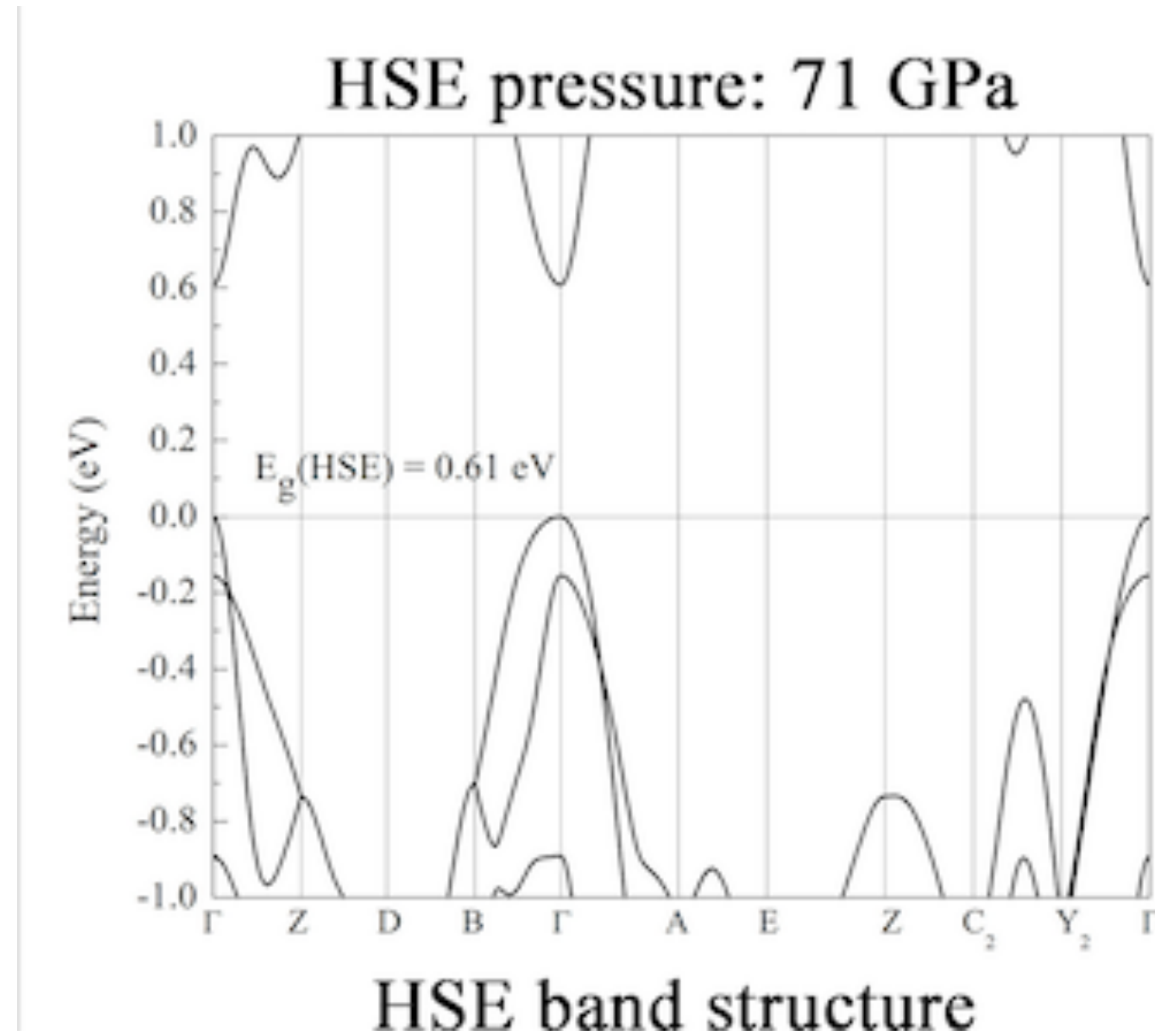
10^3

10^4

10^5

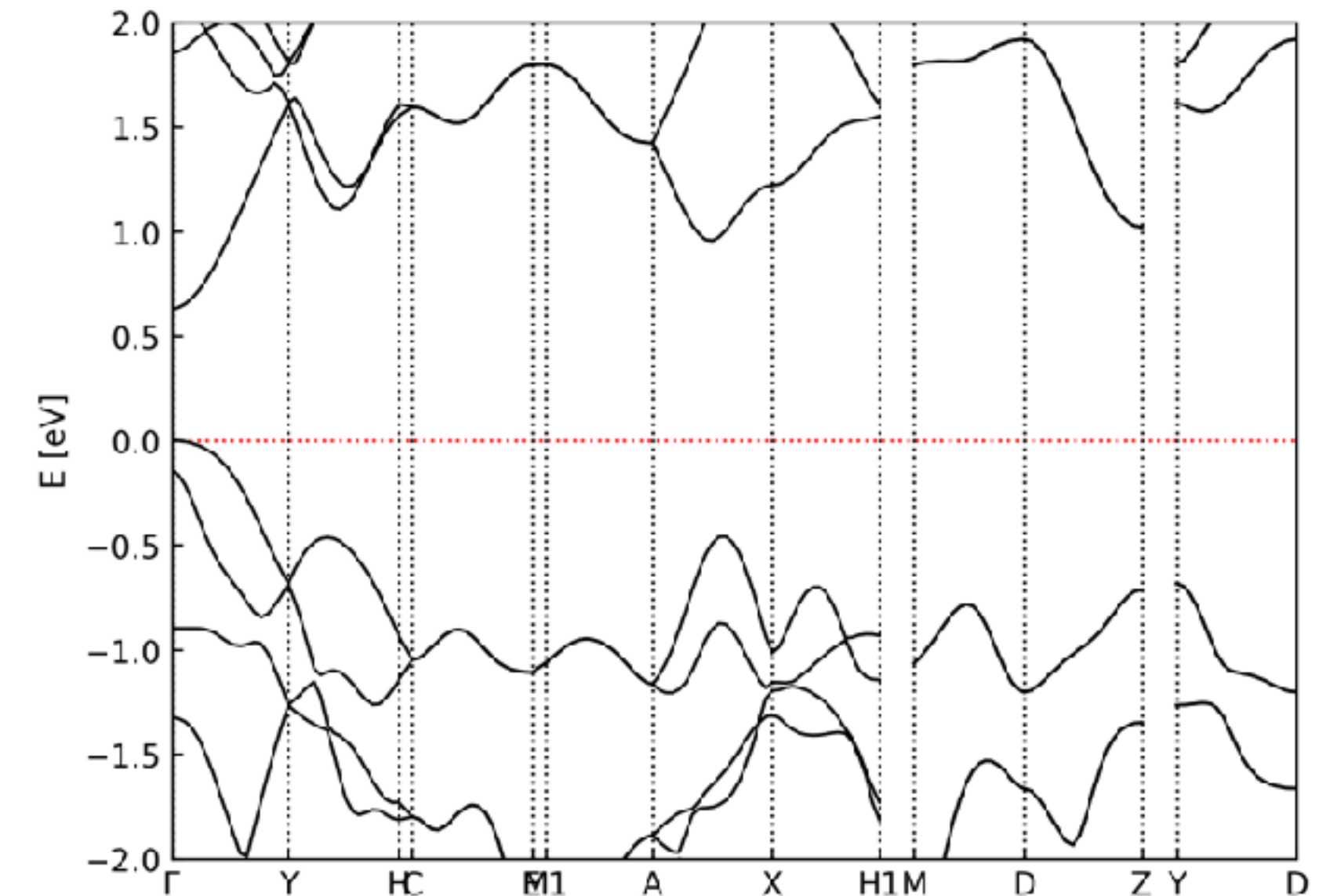
No. MPI threads

Bandstructures in reasonable time



VASP:

4 nodes, 16 cores per node, 72h walltime



FHI-aims:

2 nodes, 32 cores per node, 1.5h walltime

How does it work?

Structures: geometry.in

```
>>> ase convert POSCAR geometry.in
```

How does it work?

CLI: CLIMS uses ASE

```
>>> pip install clims
>>> clims-configure --species-path \
    ~/local/FHIaims/species_defaults/defaults_2020/
>>> clims-prepare-run --species light --relax
```

How does it work?

Calculation parameters: `control.in`

```
xc                pbe
relativistic      atomic_zora scalar
k_grid            8 8 8
relax_geometry    trm 5e-3
relax_unit_cell   full

# plus more input for the basissets
...
```

How does it work?

Calculation parameters

```
xc
relativistic
k_grid
relax_geometry
relax_unit_cell

# plus more input
...
```

```
#####
# FHI-aims file: control.in
# Created using the Atomic Simulation Environment (ASE)
# Fri Apr 19 16:55:22 2024
#####
xc                pbe
relativistic      atomic_zora scalar
k_grid            8 8 8
relax_geometry    trm 5e-3
relax_unit_cell   full
#####

#####
#
# FHI-aims code project
# VB, Fritz-Haber Institut, 2009
#
# Suggested "light" defaults for Pt atom (to be pasted into control.in file)
# Be sure to double-check any results obtained with these settings for post-processing,
# e.g., with the "tight" defaults and larger basis sets.
#####
species          Pt
#   global species definitions
nucleus           78
mass              195.084
#
l_hartree         4
#
cut_pot           3.5 1.5 1.0
basis_dep_cutoff  1e-4
#
radial_base       72 5.0
radial_multiplier 1
angular_grids specified
division          0.4222 50
division          0.9557 110
division          1.2477 194
division          1.5393 302
#
division          1.9382 434
#
division          2.0887 590
#
division          2.1534 770
#
division          2.2208 974
#
division          2.6985 1202
#
outer_grid        974
outer_grid        302
#####

#
# Definition of "minimal" basis
#
#####
```

atomic_zora scalar

How does it work?

CLI: CLIMS

```
>>> clims-prepare-run --species light --bands --dos -hse06
```

How does it work?

Calculation parameters: control.in

```
xc hse06 0.11
hse_unit bohr
hybrid_xc_coeff 0.25
relativistic atomic_zora scalar
k_grid 9 9 9
exx_band_structure_version 1
output band 0.00000 0.00000 0.00000 0.00000 0.00000 0.50000 18 G Y
output band 0.00000 0.00000 0.50000 -0.46222 0.00000 0.54058 18 Y H
output band -0.46222 0.00000 0.54058 -0.50000 0.00000 0.50000 2 H C
output band -0.50000 0.00000 0.50000 -0.50000 0.50000 0.50000 19 C E
output band -0.50000 0.50000 0.50000 -0.53778 0.50000 0.45942 2 E M1
output band -0.53778 0.50000 0.45942 -0.50000 0.50000 0.00000 17 M1 A
output band -0.50000 0.50000 0.00000 -0.50000 0.00000 0.00000 19 A X
output band -0.50000 0.00000 0.00000 -0.53778 0.00000 0.45942 17 X H1
output band -0.46222 0.50000 0.54058 0.00000 0.50000 0.50000 18 M D
output band 0.00000 0.50000 0.50000 0.00000 0.50000 0.00000 18 D Z
output band 0.00000 0.00000 0.50000 0.00000 0.50000 0.50000 19 Y D

# plus more input for the basissets
```

How does it work?

Run the job

```
#!/bin/bash -l
```

```
ulimit -s unlimited
```

```
export OMP_NUM_THREADS=1
```

```
srun /path/to/aims/build/aims.231208.scalapack.mpi.x
```

Final words



✓ **FHI-aims is easy to install** on tetralith/dardel → contact me

✓ **we have an FHI-aims license** → Florian Trybel

⚙ **we could do a small code tutorial?** → contact me

References



- [1] V. Blum *et al.*, Comput Phys Commun **180**, 2175 (2009).
- [2] S. V. Levchenko *et al.*, Comput Phys Commun **192**, 60 (2015).
- [3] A. C. Ihrig *et al.*, New J Phys **17**, 093020 (2015).
- [4] S. Kokott *et al.*, *arXiv* 2403.10343 (2024)

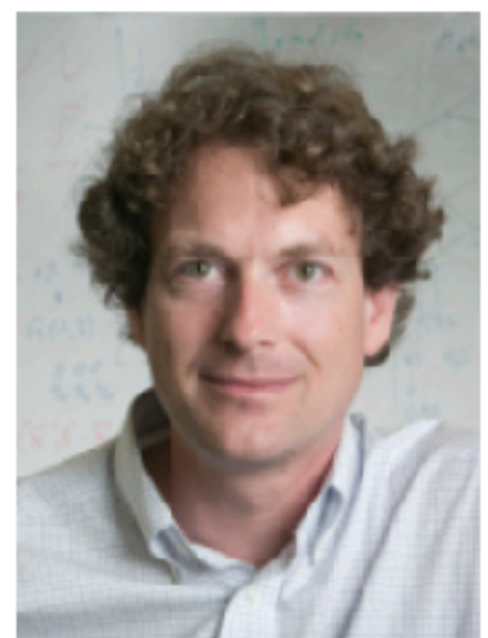
Thanks!

More info, code tutorials, etc:

fhi-aims.org



Sebastian Kokott
FHI / MSIP e.V.



Volker Blum
Duke University